

REVIEW PAPER

Plant micro- and nanomechanics: experimental techniques for plant cell-wall analysis

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Abstract

In the last few decades, micro- and nanomechanical methods have become increasingly important analytical techniques to gain deeper insight into the nanostructure and mechanical design of plant cell walls. The objective of this article is to review the most common micro- and nanomechanical approaches that are utilized to study primary and secondary cell walls from a biomechanics perspective. In light of their quite disparate functions, the common and opposing structural features of primary and secondary cell walls are reviewed briefly. A significant part of the article is devoted to an overview of the methodological aspects of the mechanical characterization techniques with a particular focus on new developments and advancements in the field of nanomechanics. This is followed and complemented by a review of numerous studies on the mechanical role of cellulose fibrils and the various matrix components as well as the polymer interactions in the context of primary and secondary cell-wall function.

Key words: Atomic force microscopy, fibre–matrix interactions, micromechanics, nanoindentation, plant biomechanics, primary cell walls, secondary cell walls.

Introduction

Plant micro- and nanomechanics are subdisciplines of plant biomechanics, which primarily address the cell and cell-wall level of the plant body (Niklas & Spatz, 2012). Due to the hierarchical structure of plants (Speck & Burgert, 2011; Gibson, 2012), research progress in this field has a great impact on plant biomechanics in general, as properties and features at this scale inevitably influence the macroscopic appearance and performance of the plant body (Fratzl and Weinkamer, 2007). In recent years, there has been a tremendous increase in mechanical characterization studies at the cell and cell-wall level dealing with various issues of structure–property and structure–function relationships, which have been reviewed by various authors (Geitmann, 2006; Burgert & Dunlop, 2011; Cosgrove & Jarvis, 2012; Eder *et al.*, 2013; Kasas *et al.*, 2013). On the one hand, this has been due to rapid methodological developments in this field that allow more precise determination of mechanical properties, in particular at the nanoscale

of the biological systems. On the other hand, cell and cell-wall mechanics are increasingly considered to be highly relevant as we gain a deeper understanding of how mechanics affects growth processes and how mechanical properties are sensed, controlled, and tuned by the plant (Cosgrove, 1993, 2005; Burgert, 2006; Telewski, 2006; Niklas, 2009). This knowledge gain substantially affects matters of plant growth and morphology as well as closing a feedback loop in terms of the biosynthesis and structure of cell-wall components. In this context, cell and cell-wall mechanics have become an important characterization tool for evaluating the impact of genetic modifications regarding effects on plant structure and growth processes, material performance, and mechanical stability of crops (Ryden *et al.*, 2003; Pena *et al.*, 2004; Bjurhager *et al.*, 2008; Bjurhager *et al.*, 2010; Hoenicka *et al.*, 2012).

Besides a tight interplay of plant biomechanics and plant morphology, cell and cell-wall mechanics are highly relevant

in various fields of plant material utilization. For instance, mechanical characterization techniques provide highly valuable information about material properties, such as fibre performance in pulp and paper (Jayne, 1959; Page *et al.*, 1971; Kersevege, 1973; Page & El-Hosseiny, 1983; Groom *et al.*, 2002a). Furthermore, the increased knowledge offers the potential to transfer principles and mechanisms evolved by nature to the design of bio-inspired materials (Sidorenko *et al.*, 2007; Dunlop & Fratzl, 2010; Martone *et al.*, 2010; Paris *et al.*, 2010).

Surprisingly, research on primary and secondary cell walls in recent decades has been conducted largely by separate communities that have rarely interacted. While primary cell-wall research has been closely related to physiological aspects due to the interplay with the living cell, research on secondary cell walls has focused on material aspects of predominately dead cells. In recent years, however, the research communities have come closer together, due in part to our increase in understanding of how genes regulate biosynthesis processes and determine the structure, chemistry, and properties of cell walls. Additionally, we see a bridging function by micro- and nanomechanical approaches, which help to identify common principles of cell-wall assembly and the resulting structure–function relationships (Burgert, 2006; Cosgrove & Jarvis, 2012). In this review, we aim to provide an overview of the basic principles of micro- and nanomechanical techniques that are applied to both primary and secondary cell-wall analysis and give examples of how these methods are used to unravel the underlying structure–property relationships of cells and cell walls in view of their specific functions.

While there has been great progress in the field of cell-wall modelling (Salmén & de Ruvo, 1985; Perez *et al.*, 1996; Yamamoto & Kojima, 2002; Besombes & Mazeau, 2005; Hofstetter *et al.*, 2005; Hanus & Mazeau, 2006; Bader *et al.*, 2011; de Borst *et al.*, 2012) and in combined approaches in which experimental data and modelling complement each other, here we focus on the experimental side and review the tremendous knowledge gain exclusively in this field. This comprises mechanical analysis of primary and secondary cell walls of various cell types, which are briefly introduced in terms of general mechanical constraints and specific functions.

Mechanical functions of primary and secondary cell walls

Micro- and nanomechanical characterization needs to thoroughly consider the functional context of the investigated plant material as this is crucial in order to gain insight into the underlying structure–function relationships. The main mechanical functions that primary cell walls need to fulfil are to provide sufficient stiffness and strength to the cell but at the same time to allow for cell growth, as well as enabling reversible changes of cell size and shape with regard to pre-stressing and organ movements (Cosgrove, 2005; Martone *et al.*, 2010). Primary cell walls have to be considered together with turgor pressure regulated by the living cell to attain stiffness in an

‘inflated’ state. With regard to the specific mechanical properties of the primary cell wall, it is fascinating to realize that the same wall has to fulfil two mutually exclusive mechanical requirements. In order to allow growth processes, primary cell walls need to be plastically deformable, whereas to provide mechanical stability or allow reversible movements such as stomata opening and closure, the primary cell wall needs to be entirely elastic (Williams & Bennett, 1982; Proseus *et al.*, 1999; Boyer, 2001; Cosgrove, 2001, 2005; Forterre *et al.*, 2005; Roelfsema & Hedrich, 2005; Moran, 2007). The mechanism by which the cell wall can ‘switch’ between elastic and plastic deformability by alterations of biomacromolecule interactions is still debated (Fry *et al.*, 1992; Cosgrove, 2000, 2005; Schopfer, 2001; Proseus & Boyer, 2006).

At a first glance, secondary cell walls appear to be by far the more pre-determined in their functionality. The rigid network of parallel aligned cellulose fibrils and matrix substances provides mechanical stability, even to dead cells (Cave, 1968; Mark & Gillis, 1970, 1973). Beyond this fundamental requirement, one can see a wide variability in structure and composition of secondary cell walls that mirrors the different functions of cell and tissue types accompanied by different mechanical property profiles (Donaldson, 2001; Wegst & Ashby, 2004; Donaldson, 2008; Eder *et al.*, 2009; Eder & Burgert, 2010). These profiles include a vast variability in terms of material stiffness, toughness and strength as well as the capability to generate both tensile and compressive mechanical stresses (Yamamoto, 1998; Lichtenegger *et al.*, 1999; Reiterer *et al.*, 1999; Farber *et al.*, 2001; Burgert *et al.*, 2007; Goswami *et al.*, 2008; Burgert & Fratzl, 2009; Clair *et al.*, 2011; Eder *et al.*, 2013). The important mechanical requirements that need to be fulfilled by primary and secondary cell walls are illustrated in Fig. 1.

Brief overview of structure and chemistry in relation to biomechanics

In view of the wide range of various cell types and cell-wall compositions, it is impossible to treat the topic comprehensively. Therefore, we will focus on a simplified illustration of the underlying principles and concepts of primary and secondary cell walls. Generally speaking, both primary and secondary cell walls can be described by means of natural fibre composites consisting of a stiff fibrous phase made of cellulose fibrils composed of crystalline and amorphous regions as well as a pliant amorphous matrix comprising various biopolymers with hydrogen bonding between the two phases (Kerstens *et al.*, 2001; Fratzl *et al.*, 2004a). According to their specific functions, both cell-wall types can be distinguished by means of structural and chemical parameters. This applies to cell-wall thickness, cellulose orientation, degree of crystallization, volume fractions of cell-wall components, composition of the matrix, chemical bonding patterns, and water content (McCann & Roberts, 1991; Pauly *et al.*, 1999; Brändström, 2001; Donaldson, 2001; Somerville *et al.*, 2004; Burgert, 2006; Donaldson, 2008; Jarvis, 2009; Cosgrove & Jarvis, 2012). It is important to note that all these parameters are also highly varied within primary and secondary

cell walls, meaning that plants manifest a vast potential to regulate properties at the nano- and microscale level of the cell walls. In [Table 1](#), crucial parameters of primary and secondary cell walls, predominantly from a biomechanical perspective, are listed. Such a tabular comparison inevitably results in a simplification, which we assume to be tolerable for the benefit of clarity.

Plant materials and characterization techniques

In this review, we intend to discuss plant materials that are examined intensively in micro- and nanomechanical analyses, as well as to introduce the related characterization techniques for biomechanical studies. In principle, any plant organ could be studied with at least one of the below-mentioned methods. However, screening of the relevant literature shows that particular plant materials have been more prominently examined

either because of being used as model systems or because of having important mechanical properties for the plant body or in application ([Fig. 2](#)).

In terms of primary cell walls, basic investigations on cell-wall structure and properties as well as growth processes have been conducted mainly on hypocotyls of various plant species ([Nakahori *et al.*, 1991](#); [Cosgrove, 1993, 2011](#); [Kutschera & Kohler, 1994](#); [Ryden *et al.*, 2003](#); [Refregier *et al.*, 2004](#)). In recent years, with the technological developments in nano-mechanical studies, even meristems have become a matter of intensive studies as very local cell-wall stiffness alterations can be monitored ([Peaucelle *et al.*, 2011](#)). More applied research activities are related to the mechanical characterization of fruit parenchyma or fruit peels ([Bargel *et al.*, 2004](#); [Landahl *et al.*, 2004](#); [Matas *et al.*, 2004](#); [Bargel & Neinhuis, 2005](#)). Micro- and nanomechanical studies on secondary cell walls at the level of basic research are aimed at unravelling general principles and mechanisms of strength and deformation behaviour, including how plants achieve and control certain property profiles.

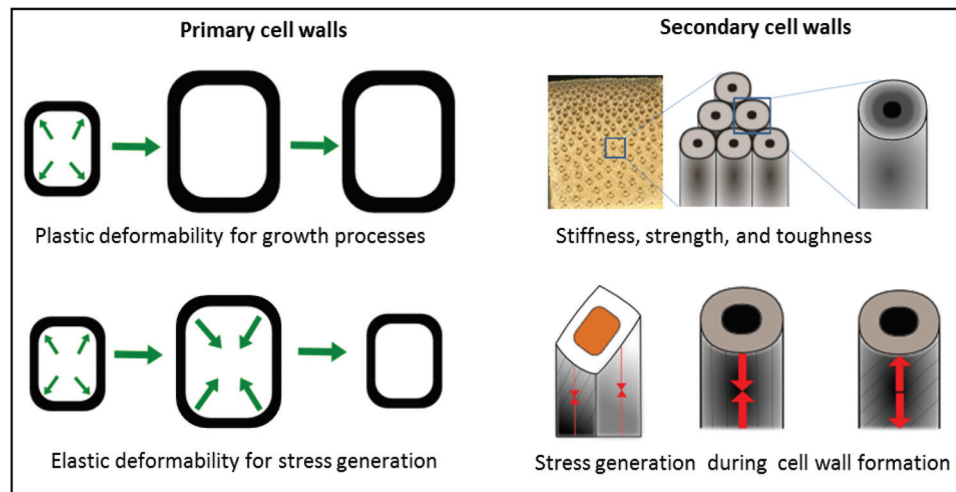


Fig. 1. Main functions and mechanical requirements of primary and secondary cell walls.

Table 1. Tabular comparison of general features of primary and secondary cell walls accompanied by cell-wall illustrations

The schematic drawing of the primary cell wall is adapted from a new cell-wall model by [Park & Cosgrove \(2012b\)](#)

Feature	Primary cell walls	Secondary cell walls
Illustration		
Cell wall	Thin, ongoing property alterations by the cell	Thick, multi-lamellar, fully differentiated structure
Polymer network	Flexible network, ongoing modification processes	Rigid network, interlocked status
Cellulose	Low content, reorientation possible, orientation more variable	High content, densely packed, strictly parallel orientation
Matrix	Predominately hemicelluloses, pectin, and structural proteins	Predominately hemicelluloses and lignin
Water interactions	Highly hydrophilic, hydrogel character	More hydrophobic when lignified
Fibre–matrix interactions—uncertainties	Widely accepted tethered cellulose–xyloglucan model recently challenged. Illustration above shows a new cell-wall model suggested by Park and Cosgrove (2012b)	Mechanical function and spatial orientation in particular of hemicellulose composition and lignin not fully understood

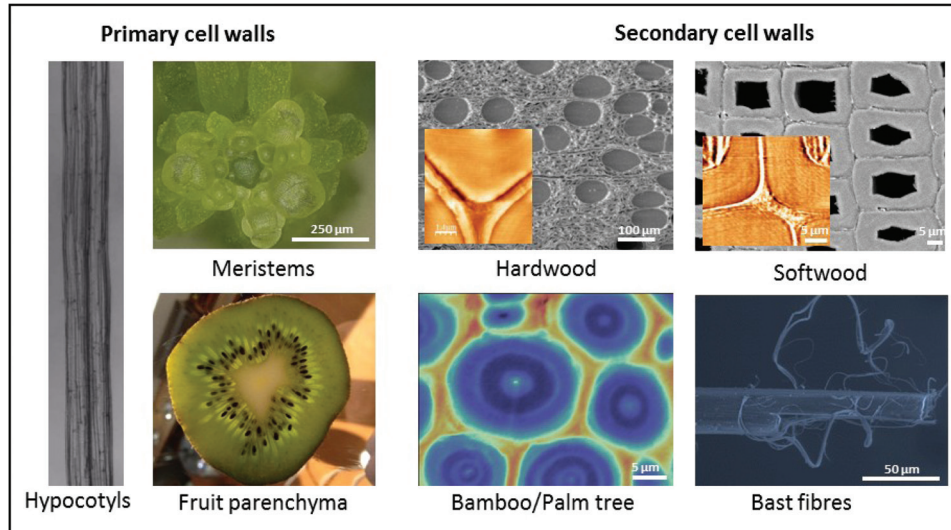


Fig. 2. Common plant materials in mechanical analysis visualized by light microscopy, scanning electron microscopy, Raman imaging and/or scanning near-field optical microscopy. Meristem image with kind permission from S. Braybrook.

In addition to establishing a general relationship between cellulose orientation and mechanical performance, understanding cellulose–matrix interactions has developed into a major focus in recent years (Reiterer *et al.*, 1999; Spatz *et al.*, 1999; Köhler & Spatz, 2002; Keckes *et al.*, 2003; Fratzl *et al.*, 2004b; Altaner & Jarvis, 2008). More applied research approaches aim at determining local mechanical properties of industrially relevant crops. A major interest is on wood fibre properties of wood species with regard to pulp and paper as well as on annually harvested bast fibres (hemp, flax) for utilization in natural fibre composites (Page & El-Hosseiny, 1983; Eichhorn *et al.*, 2001a; Bos *et al.*, 2002; Groom *et al.*, 2002a,b; Peetla *et al.*, 2006; Thygesen *et al.*, 2007). Due to the excellent mechanical performance and fast-growing capacities of bamboo and other grasses, several micro- and nanomechanical studies have been conducted in order to unravel the underlying structure–function relationships and derive mechanical characteristics for utilization purposes (Ruggeberg *et al.*, 2008; Shao *et al.*, 2010; Yu *et al.*, 2011; Wang *et al.*, 2012).

Microtensile tests

Micromechanical testing techniques address the plant level of tissues and individual cells. This includes tests on entire organs such as *Arabidopsis* hypocotyls, small tissue samples, and fibre bundles (e.g. wood segments, and fibre bundles of bamboo, palms, flax, and hemp), as well as individual cells mainly in the form of fibres and tracheids. The size and dimensions of the samples as set by plant tissue structure and variability limit the kind of loading conditions that can be applied. Accordingly, at the micromechanical level, predominantly uniaxial tensile tests are conducted, comprising common standard tests to gain information on stiffness, strength, and toughness of the plant material and time-dependent investigations to study relaxation and creep phenomena. When deriving information on cell-wall properties from mechanical tests at the tissue level, one needs to consider that a multitude of additional parameters affect the

obtained data, such as tissue density, variability of the tissue, cell length, cell–cell interactions, and turgor pressure.

For primary cell walls, the most widely utilized micro-mechanical test is a creep test in which a defined weight is attached to the tissue and the elongation is recorded over time. Based on creep tests, important information on primary cell-wall architecture and cell-wall loosening mechanisms during cell growth have been gained (Cleland, 1971; Richmond *et al.*, 1980; Suslov & Verbelen, 2006; Cosgrove, 2011). However, it needs to be emphasized that there are various constraints on what can be inferred from creep experiments and uniaxial tensile tests concerning growth processes and turgor-driven expansion, respectively (Cosgrove, 1993). Rather recently, standard tensile tests and cyclic loading tests have been applied to study the impact of genetic modification treatments on mechanical properties, in particular of dark-grown hypocotyls of *Arabidopsis thaliana* (Ryden *et al.*, 2003; Pena *et al.*, 2004; Cavalier *et al.*, 2008; Abasolo *et al.*, 2009). The hypocotyls are fixed in a microtensile tester and a uniaxial tensile stress is applied until final rupture of the specimen occurs. Methodologically, besides maintaining a sufficient level of humidity of the specimen, fixation of the hypocotyls is crucial, as the fragile and turgorized samples do not allow classical clamping, which makes a gluing process more favourable. The applied forces are recorded with load cells of small capacity, and the elongation can be derived from the machine path. However, it is always recommendable and sometimes mandatory to use optical systems such as video extensometry to record the strain more precisely and avoid measuring errors, for instance due to sample slippage in the clamps. In order to calculate the stress based on the force measurement, the cross-sectional area of the hypocotyl needs to be considered. A rather fast method is to calculate it from the diameter of the almost circular hypocotyl. However, one has to be aware that this procedure results in a stress–strain diagram for the entire hypocotyl and rules out structural differences at the tissue level such as cell size and shape, as well as cell-wall thickness.

Further parameters that influence the testing protocol and can lead to a large scatter of data and prevent direct comparisons between different studies are the influence of hypocotyl age, the germination conditions, the culture medium and the segment of the hypocotyl that has been tested.

In terms of secondary cell walls, one can state that the micro-tensile testing is a bit more determined than for primary cell walls, because the specimens are less fragile and usually dead, which rules out the influence of turgor pressure. However, in particular, specimen size can have an underestimated effect on the mechanical response of the tested plant material (Navi *et al.*, 1995). The standard tensile testing setups are similar to the one described for primary cell walls, although certain fibre bundles can be sufficiently rigid to allow fast clamping in the machine. Single fibres are usually fixed by gluing, which can be achieved either by gluing them on supporting frames that are fixed in the tensile tester or by glue droplets that are directly mounted at the fibre ends for a ball and socket setup (Kersevage, 1973; Groom *et al.*, 2002b; Burgert *et al.*, 2003; Sedighi-Gilani *et al.*, 2005). In any case, the fibres or tissue samples need to be aligned precisely, because even small deviations from the uniaxial loading condition have a large influence on the recorded mechanical behaviour due to the occurrence of shear stresses. The cross-sectional area of the sections in terms of tissues/fibres or cell walls can be determined by light or electron microscopy studies on samples that are cut transversally after testing, or cutting can be avoided when confocal light microscopy is applied (Groom *et al.*, 2002a; Burgert *et al.*, 2005). Usually, the strain required to fracture fibres and small tissue sheets containing cells with secondary cell walls is rather small, which makes a coupling of the testing setup to an optical system for high-resolution strain measurements such as video extensometry essential. Figure 3 shows in simple illustrations the most utilized micromechanical testing setups for biomechanical studies on primary and secondary cell walls.

Unique micromechanical testing approaches are so-called combined (*in situ*) methods in which external loading

is combined with simultaneous observation of nano- and microstructural deformation. The material response at the nano- and microstructural level reveals insights into specific deformation mechanisms and crack propagation events in the specimen. In plant biomechanics in particular, uniaxial tensile tests have been combined with various load-monitoring techniques to examine deformation patterns of plants at different levels of hierarchy. Detailed structural information can be obtained at the micro- and mesoscale by combining mechanical loading with light microscopy and for higher resolution with scanning electron microscopy (Mott *et al.*, 1995; Bodner *et al.*, 1996; Badel & Perré, 1999; Fruhmann *et al.*, 2003; Thygesen *et al.*, 2007; Eder *et al.*, 2008). An *in situ* approach based on computed tomography allows the three-dimensional monitoring of deformation and fracture events at the level of cells and cell-wall layers (Nazarian *et al.*, 2005; Zauner *et al.*, 2012). For an even closer look into nanostructural deformation mechanisms at the level of polymer interactions of cell-wall components, spectroscopy and X-ray scattering techniques are utilized. *In situ* infra-red and Raman spectroscopy approaches can provide information on specific load-bearing capacities of cell-wall components, as well as collective mechanical responses of polymer assemblies (Salmén & Olsson, 1998; Eichhorn *et al.*, 2000; Akerholm & Salmén, 2001; Eichhorn *et al.*, 2001b; Akerholm & Salmén, 2003; Gierlinger *et al.*, 2006; Sturcova *et al.*, 2006; Salmén & Bergstrom, 2009). *In situ* X-ray measurements at synchrotron facilities allow detailed observations of cellulose fibril-matrix interactions, including monitoring of cellulose fibril reorientation during loading and the impact of water (Keckes *et al.*, 2003; Kamiyama *et al.*, 2005; Kölln *et al.*, 2005; Zabler *et al.*, 2010).

Nanoindentation

For many decades, traditional hardness tests (pushing a tip with a defined geometry into the surface of the sample)

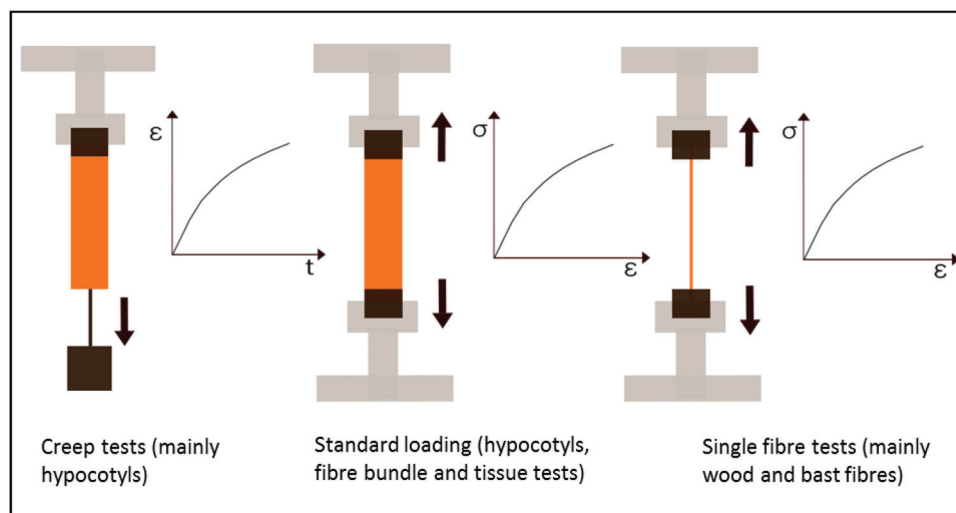


Fig. 3. Most applied micromechanical testing techniques for the characterization of primary and secondary cell walls illustrated by schematic drawings of the setup as well as of strain-time and stress-strain diagrams, respectively. (This figure is available in colour at *JXB* online.)

have been used for mechanical characterization in structural biology, including testing of cartilage, bones, soft tissues, plants, and wood in which hardness is defined as the ratio of a maximum load and the indent area (Kempson *et al.*, 1971a,b; Doyle & Walker, 1985). Nanoindentation is based on the same principle but is performed at a smaller length scale and is advanced by exactly monitoring the displacement and the loading of the indenter during the measurement. The displacement is controlled via inductance or capacitance, and for the force actuation, normally piezo elements or magnetic coils are used (Ebenstein & Pruitt, 2006; Fischer-Cripps, 2011; Oyen, 2011).

In nanoindentation, it is highly important to obtain perfectly flat surfaces in order not to distort the measurement due to surface roughness. Thus, to fulfil this requirement, an embedding material is needed (e.g. epoxy resin), which gives mechanical support during microtoming and nanoindentation. The question of penetration of the embedding material into the samples and its effect on the measurement remains unresolved. However, there is currently no conclusive evidence suggesting infiltration of the embedding material into the cell wall. For more details about sample preparation, see Konnerth *et al.* (2008) and Meng *et al.* (2013). A related limitation in nanoindentation studies towards plant biomechanics is that it is very difficult to retain the natural wet condition of the samples due to the embedding procedure. Furthermore, one has to keep in mind that only a small area/volume is tested, which is a problem in view of the vast heterogeneity of plant cell walls (de Borst *et al.*, 2012).

The main principle in nanoindentation is to calculate hardness and elastic modulus from a load-displacement (compliance) curve recorded during a local indentation. Unlike in conventional hardness tests (e.g. Vickers hardness), the size of the indent is too small for measurements with optical methods. Therefore, the area is indirectly determined from

the penetration depth together with the known geometry of the indenter (e.g. Berkovich, spherical indenter, power law indenters) (Fischer-Cripps, 2011; Oyen, 2011). Fig. 4 illustrates the testing principle and shows a typical load-displacement curve for an elastic-plastic solid.

Based on the approach by Oliver and Pharr (1992), called compliance method, the reduced modulus and the hardness are determined by analysing the load-displacement curve. For further details on the analysis of the data and the calculation of the various parameters, see Eder *et al.* (2013).

In recent last years, a tremendous effort has gone into improving the technique in order to extract further parameters such as creep compliance functions, storage modulus, and loss modulus (VanLandingham, 2003; Oyen, 2005; Fischer-Cripps, 2011; Oyen, 2011).

For data interpretation, it has to be considered that the compliance method is based on the assumption of a homogeneous and isotropic half space (Oliver & Pharr, 1992). A general restriction for biological materials is their anisotropic non-homogeneous nature. Hence, there have been huge efforts towards developing methods to reflect these constraints, such as the introduction of a holding period at peak load, continuous stiffness measurements, and multi-load indentation experiments (Ebenstein & Pruitt, 2006; Tze *et al.*, 2007; Oyen, 2011). An anisotropic indentation theory was introduced by the work of Vlassak *et al.* (2003) and applied for wood cell-wall characterization by Jager *et al.* (2011a,b). They showed that it is necessary to perform nanoindentation experiments with at least five different indentation angles (compared with the orientation of the cellulose fibrils) to determine the elastic constants with the help of an error minimization procedure. As a result of the complex behaviour of the sample under the indenter, the absolute stiffness values are smaller than expected for the longitudinal modulus (Eder *et al.*, 2013). For a comprehensive review of the mechanical modelling of cell-wall indentation, see Milani *et al.* (2013) in this issue.

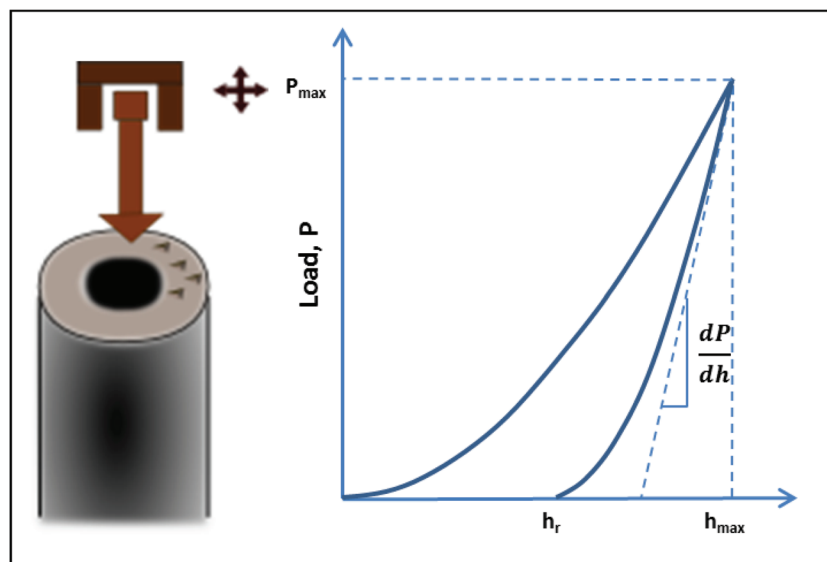


Fig. 4. Schematic of a nanoindentation test on secondary cell walls with a typical load–displacement curve for an elastic–plastic solid. (This figure is available in colour at JXB online.)

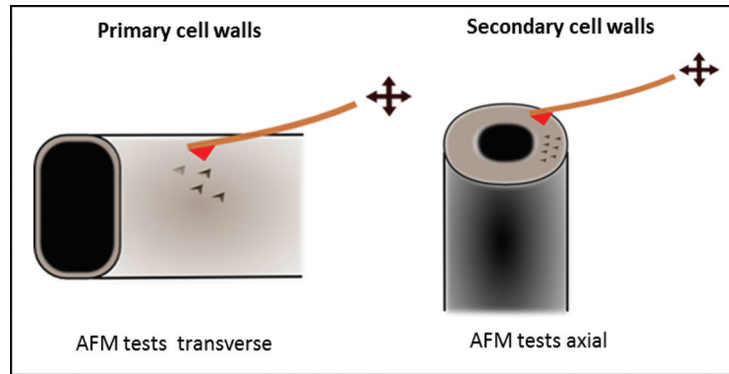


Fig. 5. Schematics of AFM in a transverse direction on primary cell walls and in the longitudinal direction on secondary cell walls. (This figure is available in colour at *JXB* online.)

Atomic force microscopy (AFM)

For measurements of mechanical properties at a smaller scale than in nanoindentation (for instance for primary cell walls), AFM has to be used. Figure 5 schematically illustrates the common loading setups for primary and secondary cell walls.

In AFM studies, a tip mounted on a cantilever spring is scanned over a sample, and the deflection of the cantilever, based on the force between the tip and the sample, is measured with the help of a photodiode. With this configuration, topography images are obtained. However, to receive mechanical properties, so-called ‘force measurements’ are necessary, where the tip is moved towards the sample in the normal direction and force–distance curves are recorded (Green *et al.*, 2002; Butt *et al.*, 2005). In contrast to nanoindentation, in AFM the results of a force measurement—cantilever deflection and piezo position—have to be converted into force and distance. First, the deflection of the photodiode has to be converted to vertical displacement and the spring constant of the cantilever is needed to calculate a force based on Hooke’s law. Additionally, a conversion for the height (change in piezo) is necessary, as it must be corrected for the deflection of the cantilever. For a detailed description of the conversion methods, see Hutter & Bechhoefer (1993), Hinterdorfer *et al.* (1996), and Butt *et al.* (2005). The main objective in using AFM for mechanical characterization is not to take force–distance curves on selected points of the sample but rather to generate images that are based on mechanical properties (force–distance curve for every pixel). Various modes for the imaging of mechanical properties are available. However, there is still much potential for improvement and new techniques. Two possible modes are resonant contact AFM and pulsed force mode. In resonant contact AFM, the tip is used as a resonator whose frequency depends on the interactions between the sample and the tip, which allows measurement of the elastic properties of the sample (Clair *et al.*, 2003). In pulsed force mode, the Z-piezo of the AFM is modulated with a sinusoidal voltage (oscillation amplitudes between 20 and 50 nm). At the lowest point of the oscillation, the tip is out of contact, and at the highest point, it reaches a deflection maximum. The outcome of pulsed force mode is force curves at high frequencies

that are recorded in real time, and high-resolution images of the mechanical properties are obtained (Krotil *et al.*, 1999).

AFM has been used intensively to characterize the molecular architecture of cell walls, such as the orientation and size of microfibrils or pore size distributions (Kirby *et al.*, 1996; Fahlen & Salmén, 2003, 2005; Yarbrough *et al.*, 2009). When interpreting the obtained data, one has to consider the fact that the stiffness values extracted from force–distance curves do not exclusively reflect the elastic properties of the cell wall (Routier-Kierzkowska *et al.*, 2012), but that indenter geometry (Bolduc *et al.*, 2006), turgor pressure (Wang *et al.*, 2004), and internal stresses also have an impact (Zamir & Taber, 2004). Another limiting factor in AFM examinations of primary cell walls is that the force that can be applied is too small to stretch the cell walls of turgid tissues. For this purpose, a so-called cellular force microscopy with the potential of applying of up to 1 mN has been developed, which allows one to indent cells until their rupture and to measure the release of the turgor pressure by changes in local stresses (Routier-Kierzkowska *et al.*, 2012).

The AFM methods for mechanical characterization introduced above are based on examining force–distance curves, but it should be also mentioned that alternative methods exist for determining mechanical properties at the nanoscale. Clair *et al.* (2003) developed resonance contact AFM for determining elastic properties of wood cells. Tetard *et al.* (2010a,b) used the so-called mode synthesizing AFM, which is based on exerting a multi-harmonic force on the substrate on the probe, which creates multiple orders coupling in frequency and allows the deduction of mechanical properties within the plant cell-wall structure. Furthermore, single-molecule force spectroscopy has been used to characterize interactions between xyloglucan molecules and a cellulose substrate (Morris *et al.*, 2004). There is ongoing research into developing new techniques for measuring mechanical properties, and many methods have simply not been utilized in plant cell-wall characterization yet (Krotil *et al.*, 1999).

Biomechanics of plant cell walls

In the following, we intend to discuss micro and nanostructural examinations on primary and secondary cell walls with

regard to the specific mechanical role of the cell-wall components, and their interaction in cell-wall assembly.

Secondary cell walls

In terms of biomechanical studies on secondary cell walls, we intend to restrict ourselves to the basic structure–property relationships, which exclude the principles of the stress-generation mechanisms mentioned and referenced above. For approaches aiming at unravelling the specific mechanical role of a cell-wall component, cellulose is probably the most easily accessible cell-wall biomacromolecule. Structure–property relationships can be derived directly due to the parallel arrangement and the partly crystalline nature of the cellulose fibrils, which allows the acquisition of detailed information on cellulose structure and formation via various methodical techniques. Hence, its crucial role in cell-wall mechanics has been shown in various studies at the tissue and fibre levels in which the tensile stiffness of the material has been related to the cellulose orientation [microfibril angle (MFA)]. The larger the MFA, the lower the stiffness of the cell wall, which enables the cell to control and adjust mechanical performance in the cellulose spinning process. Using micromechanical tests on tissue and fibres, this relationship has been shown in various studies (Page & El-Hosseiny, 1983; Lichtenegger *et al.*, 1999; Reiterer *et al.*, 1999; Saren *et al.*, 2001; Salmén & Burgert, 2009; Eder & Burgert, 2010; Eder *et al.*, 2013). *In situ* tests combining tensile straining of single wood fibres with Raman spectroscopy have directly revealed the load-bearing capacity of the cellulose fibrils, by showing a strong correlation between applied stress/strain and the nanodeformation of covalent bonds in the cellulose fibrils (Gierlinger *et al.*, 2006). In recent years, nanoindentation has also been increasingly utilized to work on secondary cell walls (Wimmer *et al.*, 1997; Gindl & Schoberl, 2004; Gindl *et al.*, 2004; Konnerth *et al.*, 2009; Wu *et al.*, 2009; Adusumalli

et al., 2010) and in particular on the mechanical impact of cellulose orientation. Figure 6 shows the correlation between MFA and cell-wall stiffness measured by nanoindentation as well as microtensile tests based on values reported in literature. Data obtained by both methods show the same trend following an increase in MFA, but in particular the decrease in stiffness measured by nanoindentation above 15° MFA is less pronounced compared with the microtensile studies. This can be explained by the different mechanical loading and testing conditions, which result in discrepancies between the methods, predominately for samples with larger MFAs, when the influence of the matrix properties becomes more relevant.

The mechanical role of cell-wall matrix components is more difficult to examine than that of cellulose, because of a less-ordered assembly and their amorphous nature. In consequence, this excludes almost entirely an unravelling of direct structure–property relationships. Hence, a common procedure is specifically to alter matrix components by chemical, enzymatic, or genetic treatments and to conduct mechanical tests to discriminate between the modified and the reference sample. The influence of individual cell-wall components in secondary cell walls has been shown by tensile tests on delignified, and cellulose or hemicellulose extracted samples (Köhler & Spatz, 2002; Konnerth *et al.*, 2010; Takeichi *et al.*, 2013). For interpretation of the obtained data, it has to be considered that the mechanical influence of the matrix components is largely affected by the cellulose orientation and the loading condition, as the influence of the matrix becomes more prominent with increasing MFA (see also Fig. 6). How this further relates to the loading condition can be seen in studies on the mechanical role of lignin. While genetic alteration in aspen trees only marginally influenced the tensile stiffness (Bjurhager *et al.*, 2010), nanoindentation tests on hardness revealed a positive correlation with the lignin content (Gindl *et al.*, 2004; de Borst *et al.*, 2012).

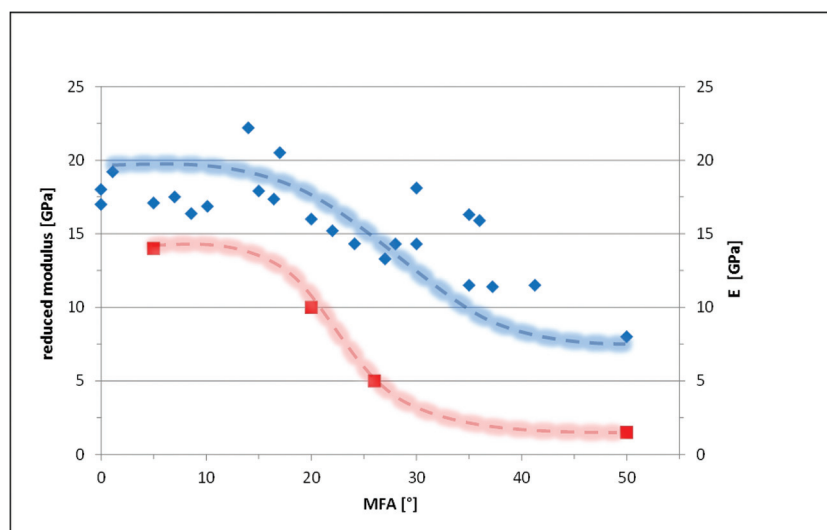


Fig. 6. Reduced moduli measured by nanoindentation plotted against microfibril angle (MFA) based on data in the literature (blue dots) of softwood cell walls (Gindl *et al.*, 2004; Tze *et al.*, 2007; Jager *et al.*, 2011a,b) and cell-wall stiffness calculated from microtensile tests plotted against MFA based on data in the literature (red dots) of spruce (Reiterer *et al.*, 1999). Dashed lines have been inserted to accent the different decreases in stiffness measured with both methods following an increase in cellulose MFA.

These interdependencies in characterizing the mechanical impact of cell-wall components underpin the fact that it is very important to gain further insight into the fibril–matrix interactions. This applies in particular to the hemicelluloses, as they are the dominating matrix components at the interface with cellulose fibrils and have a mediating function between cellulose and the other matrix components such as lignin. In this matter, *in situ* methods have contributed significantly to an advanced understanding of cell-wall deformation processes. X-ray measurements on plant tissues under external stresses showed that the orientation of cellulose fibrils can change upon external loading, in particular in wet secondary cell walls with a high MFA in the range of 30–50°. Analysis of cyclic loading behaviour indicated a passive movement or reorientation of cellulose fibrils (Köhler & Spatz, 2002; Keckes *et al.*, 2003). It is supposed that a multitude of hydrogen bonds that attach the hemicellulose chains to the cellulose surface can be opened and closed and thereby facilitate this ‘Velcro’ mechanism, which leads to a tight but highly flexible interface (Keckes *et al.*, 2003; Altaner & Jarvis, 2008). However, it needs to be mentioned that the required strains are beyond tensile deformations that can appear in woody tissues under natural conditions in living plants. Dynamic mechanical tests in combination with Fourier transform infra-red spectroscopy have allowed the division of hemicelluloses into two categories: those in close affinity to cellulose and those coupled to lignin. Additionally, this technique has provided information on the structural orientation of lignin and has thereby contributed to a more detailed model of secondary cell-wall

architecture (Salmén & Olsson, 1998; Akerholm & Salmén, 2001, 2003).

Primary cell walls

In biomechanical studies on primary cell walls, the cellulose is less easily accessible due to the low thickness of the expanding primary cell wall and the rather low cellulose content. In primary cell walls, a tilting of cellulose is supposed to be a consequence of the axial expansion of the cell walls during cell growth (Preston, 1974, 1982; Baskin, 2005). However, we currently have no *in situ* technology at hand that allows the monitoring of reorientation of cellulose fibrils during growth processes. Studies that compare the cellulose orientation before and after expansion indicate that the tilting could be less pronounced than expected (Marga *et al.*, 2005).

Mechanical tests mainly aim at unravelling the specific function of pectin and hemicelluloses in the cell-wall network, as well as of assembly-modifying substances in the growth process (e.g. expansins, enzymes). Likewise, in secondary cell walls, the mechanical function of matrix components is addressed mainly by an alteration/modification of the targeted polymer. In primary cell-wall research, genetic modifications are further advanced and enzyme treatments are more favourable due to the better accessibility of the cell-wall structure. In addition to investigations on the natural cell wall, cell-wall analogues can also be utilized in micromechanical studies. Here, the cell-wall assembly is mimicked by merging bacterial cellulose with matrix polymers obtained from plant sources. Volume fractions and the composition

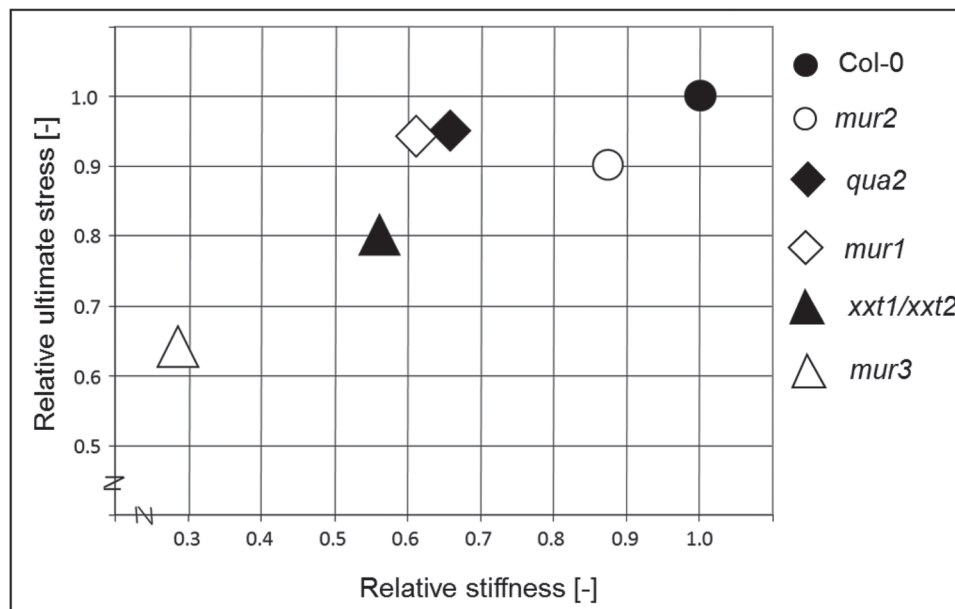


Fig. 7. Data plot of mean values of relative stiffness against relative ultimate stress of *Arabidopsis* hypocotyls to compare wild-type properties with xyloglucan and pectin mutants (4 and 6 d old); the arithmetic means are given as a percentage of the wild-type (Col-0=1.0). Detailed information on the mechanical properties (arithmetic means, standard deviation) of Col-0, *mur1*, *mur2*, and *qua2* is given in Abasolo *et al.* (2009), of Col-0 and *mur3* in Burgert (2006), and of Col-0 and *xxt1/xt2* in Cavalier *et al.* (2008). Image from *Mechanical integration of plant cells and plants*, 2011, 27–52, Micromechanics of cell walls, Burgert I, Dunlop JWC. © Springer-Verlag Berlin Heidelberg 2011. With kind permission of Springer Science+Business Media.

of these artificial cell walls can be varied and the mechanical performance analysed (Chanliaud *et al.*, 2002). Generally, the addition of matrix substances leads to a reduction in cell-wall stiffness in comparison with a pure cellulose network whereupon the type of hemicellulose has a crucial impact on the mechanical performance (Whitney *et al.*, 1995, 1999).

AFM studies on local growth zones in meristems for organ initiation have revealed the prominent influence of pectin on cell-wall stiffness, as it was shown that tissue stiffness decreased with pectin demethylesterification (Peaucelle *et al.*, 2008, 2011). AFM stiffness tomography has also been used to map the mechanical properties of *Arabidopsis* during growth. The stiffness was higher in the exponential growth phase compared with the beginning and end of the growth process (Radotic *et al.*, 2012). Creep tests in particular on hypocotyls were utilized to investigate cell-wall extension in the presence of auxin, acid conditions, and cellulose- and hemicellulose-specific enzymes as well as expansins (Kutschera & Schopfer, 1986a,b; Cleland *et al.*, 1987; Cosgrove, 1988, 1989, 1993, 1999, 2011). These investigations have largely contributed to the understanding of primary cell-wall structure and composition, as well as the mechanisms of cell-wall loosening that allow cell expansion. More recently, standard tensile and cycling loading tests on genetically modified *Arabidopsis* hypocotyls further revealed the mechanical relevance of xyloglucan composition and pectin components, as well as the binding characteristics within and between the macromolecules. It was shown that hypocotyl stiffness and strength is highly influenced by the cellulose–xyloglucan network and by pectin in terms of rhamnogalacturonan II–borate complexes (Ryden *et al.*, 2003; Pena *et al.*, 2004). A severe reduction in the pectin homogalacturonan (*qua2*) largely affected the stiffness but only marginally affected the strength of the cell walls (Abasolo *et al.*, 2009). Hypocotyls with an altered xyloglucan structure (*mur2*, *mur3*, *xxt1/xxt2*) showed a loss in stiffness and strength to different extents. Interestingly, *mur1*, which possessed a more severe alteration in pectin than in xyloglucan, closely matched the *qua2* mutant (Fig. 7).

In particular, the remaining mechanical performance of the *xxt1/xxt2* mutant could not fully be explained on the basis of the widely accepted tethered cellulose–xyloglucan model, in spite of a severe xyloglucan alteration (Cavalier *et al.*, 2008). Park & Cosgrove (2012b) suggested recently a new cell-wall model with very local connections of adjacent cellulose fibrils based on observations of creep and relaxation behaviour of the same mutant (Park & Cosgrove, 2012a) and following treatments with cellulose- and hemicellulose-specific enzymes (see also the primary cell-wall illustration in Table 1).

Conclusion and outlook

Besides structural and (bio)chemical studies, micro- and nano-mechanical techniques have become increasingly important tools to gain a deeper insight into structure–function relationships in plant materials. The obtained mechanical data contributes in a highly valuable manner to the development of

new cell-wall models, as well as a better understanding of control and adjustment of mechanical properties and cell-expansion processes during cell growth. However, one always needs to be aware of the intrinsic constraints for a comprehensive and precise experimental characterization resulting from the inhomogeneity and anisotropy of the investigated plant material as well as polymer–water interactions and pre-stresses. In consequence, the present specific limitations in the mechanical characterization techniques have to be considered when interpreting the obtained data, and complementary modelling approaches are required to gain insight into the underlying principles of plant cell-wall structure and function.

Although primary and secondary cell walls have to fulfil different mechanical functions, they are both fibre composite structures governed by the general principles and mechanisms of biomacromolecular interactions. Hence, micro- and nanomechanical characterization techniques can not only reveal basic structure–function relationships of both cell-wall types but can also build a bridge between the rather independently acting research communities. New possibilities to gain a more advanced understanding of plant cell-wall structure and biomechanics arise both from the material and the methodology side. In terms of the material, the ongoing development in particular in the field of genetic modifications will lead to a vast pool of plants with highly specific alterations of cell-wall components and bonding patterns, for both primary and secondary cell walls. In terms of methodology, we expect further progress with *in situ* techniques making them also applicable at the primary cell-wall level. An even greater impact may arise from AFM and its combination with other methods, such as tip-enhanced Raman spectroscopy and scanning near-field optical microscopy, as they have the potential to provide new insights into the basic structure–function relationships with nanoscale resolution by simultaneously collecting the topography with AFM technology and chemical information with nano-optical methods. However, these methods are still highly challenging due to a lack of fully integrated instruments, problems with introducing artefacts, and data interpretation.

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